# AD-A274 510

2

OFFICE OF NAVAL RESEARCH Grant No. N00014-91-J-1655 R&T Code 4132058---02

### TECHNICAL REPORT NO. 19

Liquid Crystalline Side-Chain Polysiloxanes with 4-Amino-4'-Stilbene-Carboxylic Ester Mesogens

bу

DTIC ELECTE JAN 04 1994

Marietta O. Bautista and Warren T. Ford

Department of Chemistry Oklahoma State University Stillwater, OK 74078

and

Randolph S. Duran and Matthias Naumann

93-31599

Department of Chemistry University of Florida Gaineville, FL 32611

Reproduction in whole or in part is permitted for any purpose of the United States Government.

This document has been approved for public release and sale; its distribution is unlimited.

## REPORT DOCUMENTATION PAGE

Form Approved OMB No 0704-0188

AGENCY USE ONLY (Leave blank)	2 REPORT DATE 12/9/93	3. REPORT TYPE AND	AND DATES COVERED		
TITLE AND SUBTITLE			S. FUNDING NUMBERS		
Liquid Crystalline Si 4-Amino-4'-Stilbene-C	de-Chain Polysilo arboxylic Ester M	canes with esogens	Grant No. N00014-91-J- 1655		
AUTHOR(S)	<del></del>				
Marietta O. Bautista, Matthias Naumann	Warren T. Ford,	R.S. Duran and			
ERFORMING ORGANIZATION NAME R. S. Duran	(S) AND ADDRESS(ES)		8. PERFORMING ORGANIZATION REPORT NUMBER		
Department of Chemist University of Florida	ry				
Gainesville, Florida	19				
PONSORING MONITORING AGENCY	NAME(S) AND ADDRESS(	:5)	10. SPONSORING / MONITORING		
Dr. Kenneth J. Wynne Code 1113PO, Office (	of the Chief of Na	val Research	AGENCY REPORT NUMBER		
800 North Quincy Stre	et	var neseuren			
Arlington, VA 22217- (703) 696-4409	-5000				
SUPPLEMENTARY NOTES					
. DISTRIBUTION / AVAILABILITY STA	TEMENT		126. DISTRIBUTION CODE		
ABSTRACT (Maximum 200 words)	a shored of polys	flovanes have been	n characterized by DSC,		
ontical microscopy	and nowder X-ray o	iffraction. The	polysiloxanes form mono		
layors at the air/way	ter interface with	molecular areas	of $30 - 35 A^2$ , and the he monolayers of CPS-1		
-2 showed good stabi	litv on water com	ared to those of .	HPS-1 and -2 as measure		
by isobaric surface	area changes as a	function of time.	Moreover, there was sion and subsequent de-		
compression below 40	mN/m for CPS-1 at	nd -2.			
4. SUBJECT TERMS			15. NUMBER OF PAGES		
			16. PRICE CODE		
7. SECURITY CLASSIFICATION 18.	SECURITY CLASSIFICATIO	19. SECURITY CLASSIF	ICATION 20. LIMITATION OF ABST		
OF REPORT	OF THIS PAGE Unclassified	OF ABSTRACT Unclassified	. (		

NSN 7540-01-280-5500

Standard Form 298 (Rev. 2-89)

#### LIQUID CRYSTALLINE SIDE-CHAIN POLYSILOXANES WITH 4-AMINO-4'-STILBENE-CARBOXYLIC ESTER MESOGENS

Marietta O. Bautista and Warren T. Ford\*
Department of Chemistry, Oklahoma State University,
Stillwater, OK 74078
and

Randolph S. Duran\* and Matthias Naumann Department of Chemistry, University of Florida, Gainesville, FL 32611

#### INTRODUCTION

Side-chain liquid crystalline polymers (SCLCPs), especially ysiloxanes, are potentially useful in electro-optical devices, <sup>1</sup> and are ily fabricated into thin films. SCLCPs with conjugated  $\pi$  electron donor 1 acceptor substituents at the ends of delocalized electron systems have ge second order nonlinear optical (NLO) coefficients,  $\chi^{(2)}$ , <sup>2</sup> commonly asured by second harmonic generation (SHG), the doubling of the quency of a laser beam. Since second order NLO polarizability requires ioncentrosymmetric structure, poled glassy polymers, and acentric igmuir-Blodgett (LB) films are promising approaches to high-efficiency G materials.

In this paper, we report the preparation, characterization, and nolayer behavior of liquid crystalline polysiloxanes having polarized benes in the side chains.

#### RESULTS AND DISCUSSION

#### ntheses

Monomers, 1 and 2, were synthesized by transesterification of the bene-methyl esters<sup>3</sup> with w-undecylenyl alcohol in the presence of Hydosilylation of the monomers with either y(hydromethyl)siloxane or a copolymer of 50% (hydromethyl)siloxane 6 (dimethyl)siloxane in dry toluene was catalyzed by chloroplatinic acid sopropyl alcohol. An equivalent amount of triethylamine was added to catalyst solution to neutralize the HCl, which otherwise would tonate the amino group of the monomers. Even after refluxing the tures for 48 hrs, a small residual 2160 cm<sup>-1</sup> band remained in the IR ctrum of the mixture, so 1-octene was added to try to consume iaining Si-H groups. The <sup>29</sup>Si NMR spectra of the final side-chain ysiloxanes showed no peak at -37 ppm for the Si-H groups. However, NMR spectra showed a small peak at 4.7 ppm for Si-H groups. HPShowed about 9% while the other polymers have about 1% of unreacted H groups present in the final structure. HPS-1 was insoluble in organic vents at room emperature but soluble in hot toluene and chloroform. PS-1 was purified by precipitation from toluene six times. CPS-1, 'S-2 and CPS-2 were soluble in organic solvents at room temperature were purified by precipitation six or seven times from chloroform into hanol. GPC analyses of the molecular weights based on polystyrene idurds and residual monomer contents are reported on Table I. GPC lyses using polystyrene standards underestimate the molecular weights CLCPs because the hydrodynamic volume of a comb-like polymer is fier than that of polystyrene of the same molecular weight.4

He I. GPC Analyses of Side-Chain Polysiloxanes

olymer	Mn	M <sub>w</sub>	M <sub>w</sub> /M <sub>n</sub>	% monomer
PS-1	6,800	9,800	1.4	<1
PS-1	5,400	8,800	1.5	<1
PS-2	19,400	37,600	1.9	2
PS-2	7,200	10,500	1.5	5

Fig. 1. Structures of monomers, homopolysiloxanes, and copolysiloxanes.

#### Phase Transitions

1

ł

Phase transitions of the monomers and the polysiloxanes measurby DSC and microscopy are reported in Table II. The phase assignmentare based on powder X-ray diffraction and polarizing optical microscop. The isotropization temperatures of all of the polymers were obtained imicroscopy. The heats of transition from the mesophase to the isotropiliquid phase were too small to observe in the DSC.

Table II. Phase Transition Temperatures of Monomers and Polysiloxani

Sample	e Phase transition temperature <sup>a</sup> , °C (ΔH,			caj n	101 <sup>-1</sup> )			
1 CPS-1 HPS-1	G	-35		146 146 (7.06) 167 (4.84)		183b 215b	i I	
2 CPS-2 HPS-2	G	-43	C C	109 (0.54) 112 (0.27) 110 (0.15)	SB			183 220h 258b

 $^a$  C = orthorhombic crystal or smectic E, G = glass, I = isotropic,  $S_A$  = smectic A, N = nematic,  $S_B$  = smectic B.  $^b$  Transition temperatures were determined by microscopy. All other data is from DSC.  $^c$  Possibly an ordered nematic.

There was no evidence of any glass transition for HPS-1 in the range -123 °C to 146 °C. This is likely due to the small volume fraction the siloxane backbone, making the glass transition too weak to detect. We strong dipole-dipole interactions between the mesogenic side chains, as long flexible methylene spacer chains, the polymer behaves semicrystalling rather than glassy. Annealing of HPS-1, even just below the isotropization to the semicrystalling of HPS-1 at 195 °C showed a diffuse peak in the wide angle region and resharp peaks in the small angle region, characteristic of a nematic phase.

CPS-I showed a weak, wide glass transition, -35 °C, and only one sophase. The X-ray pattern at 172 °C showed sharp peaks in the small tle region, indicative of smectic layers, and a diffuse peak at the wide tle region, characteristic of the SA phase. Only a fine grained texture was

served by polarizing microscopy.

HPS-2 had both a high temperature nematic phase and an ordered ise initially assigned as SB from the powder X-ray diffraction pattern at ) °C. However, this phase showed no reflections in the small angle ion, which meant the absence of layers. Possibly this mesophase is an tered nematic with hexagonal structure. Friedzon and coworkers5 orted a similar X-ray pattern and suggested that it was due to a new ise, NB, in which the mesogenic groups were packed in a hexagonal sy but without translational order in the direction of their long axes. No :roscopic texture was obtained for HPS-2.

In addition to a glass transition and a high temperature SA phase. 'S-2 had a low temperature mesophase assigned a SB structure, because powder X-ray diffraction pattern at 171 °C showed the typical sharp iks in the small angle region denoting the presence of layers and a strong irp peak in the wide angle region corresponding to 110 and 200 lections. Microscopy of CPS-2 under crossed polarized light showed a

ical fan-shaped texture for the SA phase.

Each polysiloxane exhibited a low temperature crystalline phase ntified by X-ray diffraction patterns and reported in Table II. Three up peaks in the wide angle region correspond to 110, 200 and 210 lections, which are characteristic of an orthorhombic crystal. Sharp iks in the small angle region mean the molecules are arranged in layers. ese phases might be Sg, which also gives the three reflections in the wide tle region. To distinguish a smectic E phase from an orthorhombic stal, one must use an oriented sample for X-ray diffraction studies.

Monolayers of the polysiloxanes were characterized by measurement he film pressure,  $\Pi$  - mean molecular area,  $A^2$ , isotherm. In this work, an molecular area is defined as area per mesogenic repeat unit of the ysiloxane. The room temperature isotherms in Fig. 2 were reproducible. : curves obtained for HPS-1 and CPS-1 were smooth, without any ervable discontinuities, and showed no evidence of phase isformations. Upon extrapolation of the steeply sloping linear region to o surface pressure, the intercept gives the area per molecule of a closely ked monolayer. The minimum stable areas per mesogenic polysiloxane eat unit were -23 Å<sup>2</sup> and -34 Å<sup>2</sup> for HPS-1 and CPS-1, respectively. : isotherms for HPS-2 and CPS-2 showed evidence of "phase isitions" just above 30 Å<sup>2</sup> to about 50 Å<sup>2</sup>. HPS-2 and CPS-2 gave the ne mean molecular area per repeat unit, ~ 30 Å<sup>2</sup>. Because side-chain ysiloxane films exhibited high viscosities, it was necessary to use a igmuir balance, rather than a Wilhelmy plate, to measure the surface ssure. The films can be compressed to about 40 mN/m with no apparent The copolysiloxanes showed less hysteresis than the imopolysiloxanes upon compression and subsequent expansion of the onolayers. Further compression-expansion runs showed a much approved hysteresis behavior for CPS-1 and CPS-2. Fig. 3 shows the obaric surface area of HPS-2 and CPS-2 at 15 mN/m as a function of ne. Similar stable monolayer behavior was observed at 15 mN/m for the er polysiloxanes. There was a bigger decrease of the mean molecular s of HPS-1 during the first 30 minutes than observed with CPS-1. same trend was observed with HPS-2 compared to CPS-2. The reases of mean molecular area during the first 30 minutes were much with the dimethylaminostilbene polysiloxanes, HPS-1 and CPS-1, with the hexyloxypiperidinostilbene polysiloxanes, HPS-2 and CPSlue to the greater movement of the side-chain mesogen containing the ible hexylpiperidino group.

Preliminary deposition experiments were performed on lrophobic, octadecyltrichlorosilane-treated quartz slides at room There was no deposition of the monolayers of the ysiloxanes onto a hydrophilic substrate. LB monolayers of CPS-1 e obtained with a transfer ratio of 0.8 during the first dip and a reasing transfer ratio on succeeding dips. Monolayers of CPS-2 also wed deposition, although the transfer ratio of the first dip was only 0.5. S-1 and HPS-2 did not deposit onto the hydrophobic substrate.

cnowledgement. This research was supported in part by the Office of al Research. We thank Angle Thibodeaux for instruction on the nolayer experiments.

#### REFERENCES

- McArdle, C. B., Ed. Side Chain Liquid Crystal Polymers; Blackie ar Sons: Glasgow, U.K., 1989.
- Prasad, P. N.; Williams, D. J. Introduction to Nonlinear Optical Effe in Molecules and Polymers; John Wiley & sons, Inc.: New York,
- Zhao, M.; Bautista, M.; Ford, W. T. Macromolecules, 1991, 21 84 Duran, R.; Strazielle, C. Macromolecules, 1987, 20, 2853.
- Friedzon, Y. S.; Boiko, N. I.; Shibaev, V. P.; Plate, N. A. as re; by Noel, C. In Side Chain Liquid Crystal Polymers; McArdle, C. B. Ed.; Blackie and Sons: Glasgow, U.K., 1989; p 184.

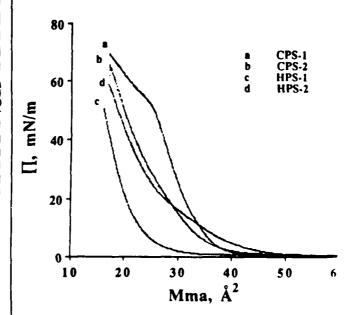


Fig. 2. Surface pressure - area isotherms of polysiloxanes. Each compression took about 1 hr.

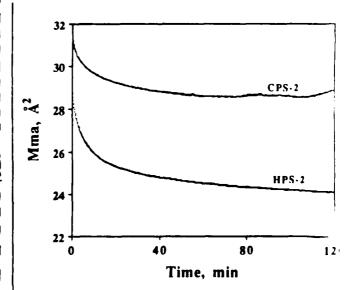


Fig. 3. Molecular area as a function of time at constant film pressure of mN/m and room temperature.

#### TECHNICAL REPORT DISTRIBUTION LIST - GENERAL

Office of Naval Research (2) Chemistry Division, Code 1113 800 North Quincy Street Arlington, Virginia 22217-5000

Dr. James S. Murday (1)
Chemistry Division, Code 6100
Naval Research Laboratory
Washington, D.C. 20375-5000

Dr. Robert Green, Director (1) Chemistry Division, Code 385 Naval Weapons Center China Lake, CA 93555-6001

Dr. Eugene C. Fischer (1) Code 2840 David Taylor Research Center Annapolis, MD 21402-5067

Dr. Elek Lindner (1) Naval Ocean Systems Center Code 52 San Diego, CA 92152-5000

Commanding Officer (1)
Naval Weapons Support Center
Dr. Bernard E. Douda
Crane, Indiana 47522-5050

Number of copies to forward

Dr. Richard W. Drisko (1)
Naval Civil Engineering
Laboratory
Code L52
Port Hueneme, CA 93043

Dr. Harold H. Singerman (1)
David Taylor Research Center
Code 283
Annapolis, MD 21402-5067

Chief of Naval Research (1)
Special Assistant for Marine
Corps Matters
Code 00MC
800 North Quincy Street
Arlington, VA 22217-5000

Defense Technical Information Center (2) Building 5, Cameron Station Alexandria, VA 22314

DING QUALITY UNSPECT IN S

Accesso:	TFo:
	(18 g) V
DTIC Urar k	
Justif C	· · · · · · · · · · · · · · · · · · ·
Ву	
Distribu	•
۸۰,	ann an is a
Dist	Special
ו מ	
	By